

Roles of resonance and dark irradiance for infrared photorefractive self-focusing and solitons in bi-polar InP:Fe

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This paper shows experimental evidence of photorefractive steady state self-focusing in InP:Fe for a wide range of intensities, at both 1.06 and 1.55 μm . To explain those results, it is shown that despite the bi-polar nature of InP:Fe where one photocarrier and one thermal carrier are to be considered, the long standing one photocarrier model for photorefractive solitons can be usefully applied. The relationship between the dark irradiance stemming out of this model and the known resonance intensity is then discussed.

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I. INTRODUCTION

Photorefractive self-focusing and spatial solitons have been up to now the object of more than a decade of growing interest, both experimentally and theoretically [1, 2, 3, 4]. During this decade, solitary waves induced by the photorefractive effect have been shown to interact with each other showing specific behaviors upon collision [5] like birth and fusion [6]. When a spatial solitary wave propagates in a photorefractive medium, it induces an optical waveguide that does not disappear instantly when the beam is turned off [7, 8], leading to optically induced optical components such as junctions [9], couplers [10] and even optically induced soliton arrays and photonic lattices [11, 12, 13].

The attraction towards photorefractive solitons and the waveguides they induce owes a lot to their potentialities in optically inducing photonic circuitry in the bulk of a material, on the one hand, and to their re-configuration properties, on the other hand. The former relies on the material property of retaining the waveguide once the beam is shut off, like, for instance, LiNbO₃ [14]. The latter, on the contrary, relies on the possibility to rapidly screen out an applied field with a laser beam, to give rise to a waveguide, and to erase this waveguide as rapidly. Up to now and to our knowledge, the characteristic build-up and erase times are on the order of the second in the widely used Sr_xBa_{1-x}Nb₂O₆ [15] or sillenite Bi₁₂TiO₂₀ [16], or up to the minute in LiNbO₃ [17], all of these experiments having been conducted at visible wavelengths to which most of the cited materials are sensitive. Shorter response times around the millisecond have been obtained in semiconductor CdZnTe [18] and

around the nanosecond for visible ultra-high intensities in Bi₁₂TiO₂₀ [19].

If one wants to target optical networking as a potential application of the dynamically reconfigurable photorefractive soliton technology, build-up and reconfiguration times below the millisecond is to be achieved ideally at infrared wavelengths compatible with optical fiber networks. Thanks to their high carrier mobility and despite their low electro-optic coefficient, photorefractive semiconductors such as InP:Fe might respond to this demand. Charge transport, in this material, has a bi-polar nature with photoexcited holes co-existing with thermally excited electrons. This yields a peculiar resonance of the photorefractive Two Wave Mixing (TWM) gain as a function of intensity [20, 21, 22, 23].

Owing to its bipolar transport character, the physics behind spatial soliton propagation in InP:Fe is more complicated than in other materials. It has already been the subject of theoretical analysis [24]. In this work, we present a series of recent investigations on photorefractive self-focusing in InP:Fe at infrared wavelengths and we propose a model that takes into account the transport of both charge carriers. Unlike in previous work [24, 25, 26], the experimental results in our crystal samples and the corresponding theoretical prediction do not show any clear evidence of an influence of the intensity resonance of the bi-polar photorefractive effect, thus indicating that photorefractive self-focusing behaves to a large extent like in the case of the well-known single-carrier model [27]. Hint is finally provided that the reason why qualitatively different behaviors are observed in different samples might be a varying oxydo-reduction state.

II. EXPERIMENTAL EVIDENCE OF THE ROLE PLAYED BY DARK IRRADIANCE

The bi-polar photorefractive resonance mechanism was first evidenced experimentally by Mainguet and Idrissi *et. al.* [20, 21] and then defined by Picoli *et. al.* [22, 23] as the mean intensity at which the Two Wave Mixing (TWM) gain is greatly enhanced. It corresponds to the intensity at which the intensity dependent hole generation rate is equal to the temperature dependent electron generation rate. The resonance intensity is thus expected to depend on the temperature and has been evidenced so. It has also been shown that this TWM enhancement corresponds to a $\pi/2$ phase shift between the index grating and the optical intensity grating, and to a corresponding enhancement of the index modulation amplitude.

As a preliminary characterization of our samples, we evidenced and measured the resonance intensity in the samples used in our experiments. The iron content of the various samples we tested varied from $8 \times 10^{16} \text{cm}^{-3}$ to $2 \times 10^{17} \text{cm}^{-3}$, as measured by Secondary Ion Mass Spectroscopy. Similar behavior were observed in all samples. The results presented in this paper were obtained with a $2 \times 10^{17} \text{cm}^{-3}$ Fe doped sample.

The resonance intensity has been measured with a standard TWM setup [20] as described in [28], for various temperatures, and particularly at 20°C , the temperature at which further self-focusing experimental results have been obtained. The measured resonance intensity at 20°C is 300mW/cm^2 at a wavelength of $1.06 \mu\text{m}$ and 3W/cm^2 at a wavelength of $1.56 \mu\text{m}$. This wavelength disparity is explained by the wavelength dependent photo-excitation cross sections.

The experiments were conducted independently at both $1.06 \mu\text{m}$ and $1.56 \mu\text{m}$ wavelengths, the latter because it corresponds to a telecom wavelength and the former for InP:Fe large sensitivity to it. The $1.06 \mu\text{m}$ wavelength is delivered by a solid state diode pumped Nd:YAG laser whereas the $1.56 \mu\text{m}$ beam is produced by a laser diode.

The beam is focused down to a $30 \mu\text{m}$ waist onto the entrance face of a temperature regulated InP:Fe sample. The beam propagates along the $\langle \bar{1}10 \rangle$ direction and is linearly polarised along $\langle 110 \rangle$ with an electric field of 10kV/cm applied along $\langle 001 \rangle$. The crystal dimensions are $5 \times 5 \times 12 \text{mm}^3$ along the directions $\langle 001 \rangle \langle 110 \rangle$ and $\langle \bar{1}10 \rangle$, respectively.

The steady state beam profile on the sample output face is observed through a microscope objective and a silicon CCD camera. The camera is sensitive enough for the $1.06 \mu\text{m}$ wavelength but requires to be equipped with a luminescent phosphorus converter to achieve $1.56 \mu\text{m}$ observations. The observed image is then gamma corrected to account for the conversion non-linearity.

The output beam profile is observed at steady state for various intensities, at a temperature of 20°C . The best result giving the strongest focusing is shown in figure 1(b) for both $1.06 \mu\text{m}$ and $1.56 \mu\text{m}$ wavelengths. Varying the input intensity reveals that the steady-state self-

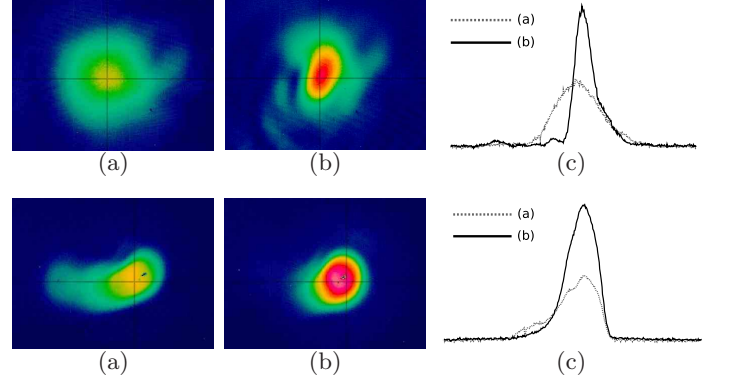


FIG. 1: (Color online) Steady state output beam profile for a peak intensity of 190mW/cm^2 at $1.06 \mu\text{m}$ (top) and a peak intensity of 8.7W/cm^2 at $1.56 \mu\text{m}$ (bottom), for no applied field (a) and a 10kV/cm horizontally applied field (b). The images total width is $200 \mu\text{m}$. The non Gaussian shape of the $1.56 \mu\text{m}$ beam in absence of the applied field is due to the multimode character of the laser itself. The curves (c) are the horizontal beam profiles (a.u.) at beam maximum.

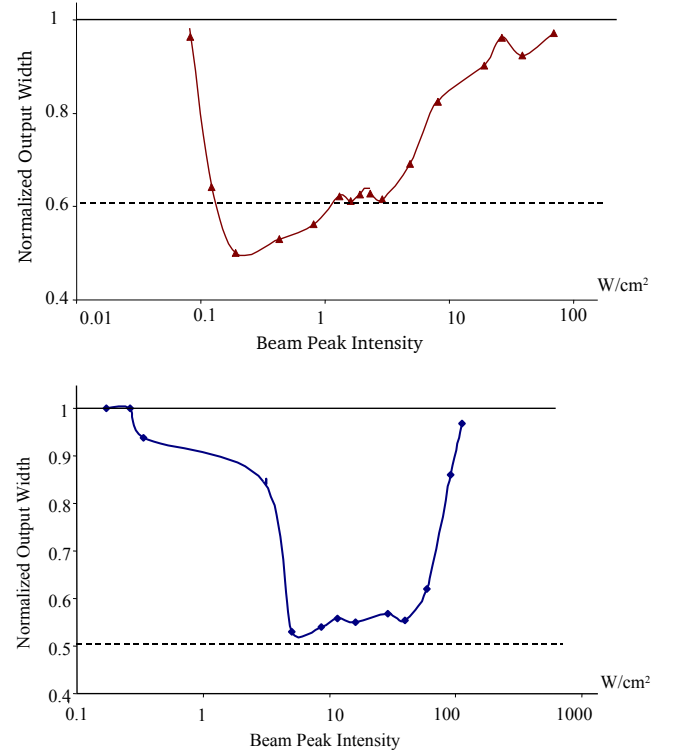


FIG. 2: Output beam width as a function of input beam peak intensity at $1.06 \mu\text{m}$ (top) and $1.56 \mu\text{m}$ (bottom) wavelengths. The beam diameter is normalized to the linearly diffracted output beam diameter. The horizontal dotted line represents an output beam as wide as the input one, suggesting a spatial soliton like propagation.

focusing phenomenon vanishes at low and high intensities and reaches an optimum focusing for an intermediate intensity, when the output beam width reaches a minimum.

This is quantified on figure 2 where the normalized horizontal output beam diameter is plotted as a function of the input beam peak intensity for an applied field of 10kV/cm. Linear diffraction is observed at steady state for small and large intensities. A minimum output beam diameter is reached for an intermediate intensity of approximately 210mW/cm^2 at $1.06\mu\text{m}$ and 7.5W/cm^2 at $1.56\mu\text{m}$. A noticeable characteristic of both curves in figure 2 is that the upside down bell-shape is *attracted* by the dotted line when it approaches it, *i.e.* when the output beam diameter approaches the input one. This evidences a domain of stability of the beam width (from 1 to 4W/cm^2 at $1.06\mu\text{m}$ and from 5 to 50W/cm^2 at $1.56\mu\text{m}$). The horizontal beam width associated to these domains roughly corresponds to the input beam width. This may suggest that the beam evolution is *attracted* towards a solitary wave solution

We note that the width stability intensity range is much wider at $1.56\mu\text{m}$ than at $1.06\mu\text{m}$: this is due to the fact that in the former case, the intensity that full-fills soliton propagation conditions is much nearer to the minimum of the reverse bell-shaped curve. Furthermore, one has to keep in mind that the beam intensity decreases with propagation, owing to absorption, — 2cm^{-1} at $1.06\mu\text{m}$ and 0.5cm^{-1} at $1.56\mu\text{m}$ — so that the soliton propagation conditions cannot be fulfilled precisely all the way through the crystal, thus strengthening the need for stability if a spatial soliton is to propagate.

As in many other materials, self-focusing in InP:Fe comes together with beam bending, as is evidenced by figure 1 and particularly 1(c). The steady-state beam shift at the output face with respect to the propagation axis, along the same horizontal direction as the applied field, is reported on figure 3. We observe that beam bending is negligible for sufficiently low and high intensities, that is, when self-focusing is absent. Furthermore, the maximum beam shift is obtained precisely at the intensity at which self-focusing is most efficient.

The curves reported on figure 2 resemble the photorefractive soliton existence curve —giving the soliton width as a function of the beam peak intensity to dark irradiance ratio— as predicted earlier [27, 29] in the framework of a single-photo-carrier model. The dark irradiance plays a central role in this theory since it is the ratio of the beam peak intensity to the dark irradiance which determines the steady state soliton width. The dark irradiance is the equivalent uniform optical irradiance which should be shone on the crystal to generate as much photocarriers as thermal excitation does. In other words, shining the dark irradiance on a given sample doubles its conductivity.

If we assume that this one-photo-carrier model — which will be named model I in the following— describes our experiments in InP:Fe, we can use its predictions to deduce the dark irradiance in InP:Fe at both the wave-

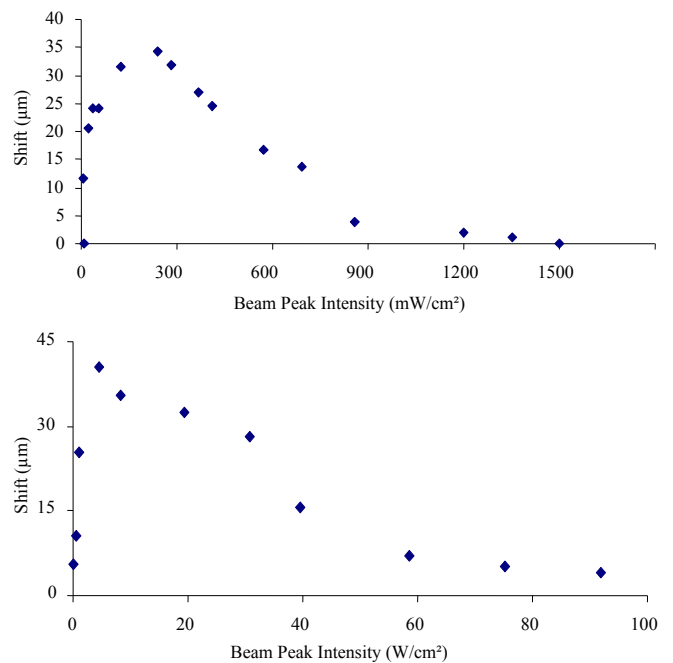


FIG. 3: Output beam horizontal shift as a function of input beam peak intensity at $1.06\mu\text{m}$ (top) and $1.56\mu\text{m}$ (bottom) wavelengths.

lengths we have used. According to it, the minimum width soliton which corresponds to the maximum self-focusing effect is obtained for a beam peak intensity equal to 3 times the dark irradiance, as confirmed recently [30] in a (1+1)D model. Using this hypothesis, the dark irradiance in our sample can then be evaluated, from figure 2, to approximately 70mW/cm^2 for the $1.06\mu\text{m}$ wavelength and 2.5W/cm^2 for $1.56\mu\text{m}$, at a temperature of 20°C ; the accuracy of the value measured at $1.06\mu\text{m}$ is low because of InP:Fe strong absorption at this wavelength. Let us note here that, in both cases, the estimated dark irradiance and the resonance intensity are on the same order of magnitude.

At this stage, one may wonder why model I should describe photorefractivity in InP:Fe since both holes and electrons are known to play a significant role [22, 23] in this semiconductor —holes are photo-excited and electrons are thermally generated— whereas model I assumes only one carrier, both photo and thermally excited. One may hastily think that electrons can simply be neglected in our experiments. This is however not the case since both the dark irradiance and the resonance intensity are included in our measurement range: they respectively correspond to the equality of the concentrations on the one hand and of the generation rates on the other hand, of holes and electrons.

Therefore, since model I assumptions do not match with our experiments, we now need to ascertain by an independent measurement that the values of the dark irradiance we have deduced from figures 2 and 3 are in-

deed correct. We have thus shone a uniform intensity on our InP:Fe sample and measured its photo-conductivity. This allowed us to deduce a $50\text{mW}/\text{cm}^2$ dark irradiance at $1.06\mu\text{m}$ and a $2\text{W}/\text{cm}^2$ one at $1.56\mu\text{m}$, confirming the previous evaluation at this wavelength. The slight discrepancy is accounted for by the relative precision with which figure 2 allows the determination of the dark irradiance on the one hand and on the other hand by the difficulty to shine a strong and uniform light on the whole sample for the independent measurement.

These recent experimental results evidence the fact that the dark irradiance plays a crucial role in photorefractive self-focusing in InP:Fe, as it does in previously developed models, despite the greater complexity of the photorefractive mechanisms in InP:Fe. However, as previously explained, model I does not allow to explain our observations, nor do previously published ones [24, 25, 26], as will be detailed later on. Therefore, we will now tackle the theoretical problem of showing how and why the dark irradiance plays a significant role in InP:Fe.

III. SPACE CHARGE FIELD AND DARK IRRADIANCE

Let us start with the general two-carrier band transport model [20, 24] reduced to one dimension x , which we will consider at steady state in the following. It is described by the set of equations below:

$$\frac{\partial E}{\partial x} = \frac{e}{\epsilon}(N_D - N_A + p - n - n_T), \quad (1a)$$

$$j_n = e\mu_n nE + \mu_n k_B T \frac{\partial n}{\partial x}, \quad (1b)$$

$$j_p = e\mu_p pE - \mu_p k_B T \frac{\partial p}{\partial x}, \quad (1c)$$

$$\frac{\partial n}{\partial t} = e_n n_T - c_n n p_T + \frac{1}{e} \frac{\partial j_n}{\partial x}, \quad (1d)$$

$$\frac{\partial p}{\partial t} = e_p p_T - c_p p n_T - \frac{1}{e} \frac{\partial j_p}{\partial x}, \quad (1e)$$

$$\frac{\partial n_T}{\partial t} = e_p p_T - e_n n_T - c_p p n_T + c_n n p_T, \quad (1f)$$

$$N_T = n_T + p_T, \quad (1g)$$

where E is the electric field, n and p are the electron and hole densities in the respective conduction and valence bands, $n_T = [\text{Fe}^{2+}]$ is the density of ionized occupied traps, $p_T = [\text{Fe}^{3+}]$ is the density of neutral unoccupied traps, j_n and j_p are the electron and hole current densities, respectively. N_T , N_D and N_A are the densities of iron atoms, the shallow donors and the shallow acceptors, respectively. The charge mobilities are given by μ_n for electrons and μ_p for holes, the electron and hole recombination rate are respectively c_n and c_p , T is the temperature and k_b is the Boltzmann constant. The dielectric permittivity is given by ϵ while e is the elementary charge. The electron and hole generation rates

e_n and e_p depend on both thermal and optical emission as described by

$$e_n = e_n^{th} + \sigma_n I \approx e_n^{th}, \quad (2a)$$

$$e_p = e_p^{th} + \sigma_p I \approx \sigma_p I \quad (2b)$$

where the thermal contribution to the emission rate coefficient is e^{th} and the optical cross section of the carriers is given by σ , I is the spatially dependent intensity of light. The approximations in both above equations are valid in the case of InP:Fe as stated above, with thermally excited electrons and photo-excited holes. The need for a large applied electric field leads us to neglect carrier diffusion when compared to drift, diffusion which is accounted for by all the terms which contain $k_B T$.

Solving the differential system (1) requires defining the boundary conditions. To that aim, we will consider that the input beam is small with respect to the sample dimensions and thus that its influence is limited to its surroundings. Therefore, the boundary values for E , n , p , n_T , p_T , j_n and j_p are the values reached at steady state in the dark. As was previously done [20, 22, 23, 28], they can be deduced from system (1), setting to zero all derivatives and optical intensity. Following, if the 0 subscript identifies the boundary condition: $E_0 = V/d$ where V is the applied voltage and d the crystal width; $n_0 = e_n^{th} n_{T0} / c_n p_{T0}$ and $p_0 = e_p^{th} p_{T0} / c_p n_{T0}$ where $n_{T0} = N_D - N_A$ and $p_{T0} = N_T - n_{T0}$ are the ionised and non-ionised iron densities in the dark [22, 23]; $j_n = e\mu_n n_0 E_0$ and $j_p = e\mu_p p_0 E_0$.

The time evolution of the electric field E is determined as follows. The time derivative of Eq. (1a) is simplified using Eqs. (1d, 1e, 1f) to yield

$$\frac{\partial^2 E}{\partial t \partial x} = \frac{-1}{\epsilon} \frac{\partial (j_n + j_p)}{\partial x}. \quad (3)$$

Making use of Eqs. (1b, 1c) and integrating with respect to the space variable x , it becomes

$$\frac{\partial E}{\partial t} = \frac{-e}{\epsilon} [(\mu_n n + \mu_p p) E - (\mu_n n_0 + \mu_p p_0) E_0] \quad (4)$$

At steady state, the evolution equations (1d), (1e), (1f) and (4) reduce to

$$(\mu_n n + \mu_p p) E - (\mu_n n_0 + \mu_p p_0) E_0 = 0, \quad (5a)$$

$$e_n n_T - c_n n (N_T - n_T) + \mu_n n \frac{\partial E}{\partial x} + \mu_n E \frac{\partial n}{\partial x} = 0, \quad (5b)$$

$$e_p (N_T - n_T) - c_p p n_T - \mu_p p \frac{\partial E}{\partial x} - \mu_p E \frac{\partial p}{\partial x} = 0. \quad (5c)$$

To further simplify this differential system, the light intensity spatial variations will be assumed small enough so that the resulting free carriers spatial variations remain small, and the derivatives $\partial n / \partial x$ and $\partial p / \partial x$ can be neglected in equations (5b) and (5c). In the same spirit, we also assume that the field derivative $\partial E / \partial x$ in

the same equations can be neglected, precisely $\partial E/\partial x \ll e_n^{th} N_T/\mu_n n_0$.

The left-hand-side of Eq. (1a) becomes zero within this approximation. Light intensity and temperature will again be considered not so large so that free carriers density remain small with respect to ionised iron density: $n \ll n_T$ and $p \ll n_T$, so that equation (1a) reduces to $n_T = n_{T0}$, and consequently from Eqs. (5b) and (5c) we get $n = n_0$ and $p = (\sigma_p I p_{T0}) / (c_p n_{T0})$. The assumptions made are valid for intensities small against $c_p n_{T0}^2 / \sigma_p p_{T0}$, itself greater than the TW/cm², and at temperatures orders of magnitude above room temperature. Within these approximations, we obtain the electric field E dependency upon the local intensity I :

$$E = \frac{I_d}{I_d + I} E_0. \quad (6)$$

This expression is strictly the same as the one found before in the framework of model I [27, 29] when diffusion is neglected; only the expression of the dark irradiance I_d differs (the approximation below being valid in the case of InP:Fe):

$$I_d = \frac{e_p^{th} p_{T0}^2 c_n \mu_p + e_n^{th} n_{T0}^2 c_p \mu_n}{\sigma_p p_{T0}^2 c_n \mu_p + \sigma_n n_{T0}^2 c_p \mu_n} \approx \frac{e_n^{th} n_{T0}^2 c_p \mu_n}{\sigma_p p_{T0}^2 c_n \mu_p}. \quad (7)$$

Let us point out here that this has implications beyond the above equation: all the conclusions that have been driven, in model I, from an equation identical to (6), still hold, if the above value of I_d is accounted for. There is for instance no need to derive again the soliton existence curve since the involved equations are the same. Therefore and using the predictions model I, our observations of self-focusing in InP:Fe are partially explained. Indeed the presence of self-focusing over a broad range of intensity is predicted and no electron-hole resonance effect is observed. The beam bending is however not yet accounted for, which is one reason to further extend and refine our model.

IV. RESONANCE INTENSITY

As was said before, we felt the need to establish the theoretical analysis presented in the previous section — which we will now call model II — because we could not satisfactorily explain our observations using the theory available in the literature [24, 26] for one particular reason: this theory — which we will call model III — predicts the existence of a resonance intensity around which the self-focusing phenomenon reverses itself; while we have successfully measured the resonance intensity in our samples, we were never able to measure the predicted reversal from focusing to defocusing.

The resonance intensity around which a self-focusing behavior reversal is predicted by model III is precisely the same resonance intensity at which TWM gain is maximum. As was said before, the resonance intensity is also

the intensity at which the generation rates of holes and electrons are equal:

$$I_r = \frac{e_p^{th} p_{T0} + e_n^{th} n_{T0}}{\sigma_p p_{T0} + \sigma_n n_{T0}} \approx \frac{e_n^{th} n_{T0}}{\sigma_p p_{T0}}. \quad (8)$$

At resonance, the imaginary part of the space charge field reaches a maximum while the real part — the *in phase* component — reaches zero. Following, the real part changes sign around I_r [22, 26]. The link with self-focusing can then be thought of as follows: the in phase component of the space-charge field — the real part — is a *local* photorefractive response, responsible for self-focusing or defocusing, whereas the out-of phase one — the imaginary part — is responsible for beam bending. This phenomenological interpretation has been confirmed by an experiment [26]. Later on, analytical derivations and numerical simulations [24] came to the same predictions. As stated before, however, we were never able to reproduce the experiment, for a reason we will now try to give hints of.

First of all, in the case of InP:Fe, there is a close and fairly simple relationship between I_d and I_r :

$$\frac{I_d}{I_r} = \frac{\mu_n c_p}{\mu_p c_n} \times \frac{n_{T0}}{p_{T0}}. \quad (9)$$

We have evaluated the above ratio to 0.65 with values issued from the literature [31, 32, 33], a numerical value which is fully compatible with those we have measured at 1.56 μm , the discrepancy for the measures at 1.06 μm being accounted for by the strong absorption impairing the I_d estimation from figures 2 and 3.

The above ratio (9) evidences the ratio $[\text{Fe}^{2+}]/[\text{Fe}^{3+}]$ in the dark (n_{T0}/p_{T0}). Therefore, the oxydo-reduction state plays a significant role, as well as the doping of the samples, in the ratio of the two characteristic intensities. Therefore, a different doping but also a different $[\text{Fe}^{2+}]/[\text{Fe}^{3+}]$ ratio might be the cause of the discrepancy between our experimental results, which do not exhibit a crucial role for the resonance intensity, and those previously reported.

However, our theoretical analysis, model II, does not either bring the resonance intensity forth. A more universal model is in development in order to predict a doping dependent behavior and thus unify model II and model III.

V. CONCLUSION

In the light of recent experimental results, we have shown that despite more complex photorefractive charge transport mechanisms, steady-state photorefractive self-focusing and solitons in InP:Fe can be described by the same model as was done for the simpler one-free-carrier model, provided diffusion is neglected. We have evidenced in the process the role of the dark irradiance. Our

model does not, however, evidence any crucial role for the resonance intensity, a role which is to be the subject of future studies.

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